Effect of Zn doping on the antiferromagnetism in kagome Cu$_{4-x}$Zn$_x$(OH)$_6$FBr

Zili Feng,[1,2] Yuan Wei,[1,2] Ran Liu,[1] Dayu Yan,[1,2] Yan-Cheng Wang,[1,3] Jianlin Luo,[1,2,4] Anatoliy Senyshyn,[5]
Clarina dela Cruz,[6] Wei Yi,[7] Jia-Wei Mei,[8] Zi Yang Meng,[1,9] Youguo Shi,[1,*] and Shiliang Li[1,2,4,†]

[1]*Beijing National Laboratory for Condensed Matter Physics, Institute of Physics, Chinese Academy of Sciences, Beijing 100190, China*
[2]*School of Physical Sciences, University of Chinese Academy of Sciences, Beijing 100190, China*
[3]*School of Physical Science and Technology, China University of Mining and Technology, Xuzhou 221116, China*
[4]*Collaborative Innovation Center of Quantum Matter, Beijing 100190, China*
[5]*Heinz Maier-Leibnitz Zentrum (MLZ), Technische Universität München, Garching D-85747, Germany*
[6]*Neutron Scattering Division, Neutron Sciences Directorate, Oak Ridge National Laboratory, Oak Ridge, Tennessee 37831, USA*
[7]*Semiconductor Device Materials Group, National Institute for Materials Science, 1-1 Namiki, Tsukuba, Ibaraki 305-0044, Japan*
[8]*Institute for Quantum Science and Engineering, and Department of Physics, Southern University of Science and Technology,
Shenzhen 518055, China*
[9]*CAS Center of Excellence in Topological Quantum Computation, University of Chinese Academy of Sciences, Beijing 100190, China*

Barlowite Cu$_4$(OH)$_6$FBr shows three-dimensional (3D) long-range antiferromagnetism, which is fully suppressed in Cu$_3$Zn(OH)$_6$FBr with a kagome quantum spin liquid ground state. Here we report systematic studies on the evolution of magnetism in the Cu$_{4-x}$Zn$_x$(OH)$_6$FBr system as a function of x to bridge the two limits of Cu$_4$(OH)$_6$FBr ($x = 0$) and Cu$_3$Zn(OH)$_6$FBr ($x = 1$). Neutron-diffraction measurements reveal a hexagonal-to-orthorhombic structural change with decreasing temperature in the $x = 0$ sample. While confirming the 3D antiferromagnetic nature of low-temperature magnetism, the magnetic moments on some Cu^{2+} sites on the kagome planes are found to be vanishingly small, suggesting strong frustration already exists in barlowite. Substitution of interlayer Cu^{2+} with Zn^{2+} with gradually increasing x completely suppresses the bulk magnetic order at around $x = 0.4$ but leaves a local secondary magnetic order up to $x \sim 0.8$ with a slight decrease in its transition temperature. The high-temperature magnetic susceptibility and specific-heat measurements further suggest that the intrinsic magnetic properties of kagome spin liquid planes may already appear from $x > 0.3$ samples. Our results reveal that the Cu$_{4-x}$Zn$_x$(OH)$_6$FBr may be the long-thought experimental playground for the systematic investigations of the quantum phase transition from a long-range antiferromagnet to a topologically ordered quantum spin liquid.

I. INTRODUCTION

A quantum spin liquid (QSL) can be briefly described as a symmetric state without magnetic order emerging from strong quantum fluctuations in frustrated magnetic systems [1–3]. The quantum fluctuations are usually enhanced by geometrical frustrations of magnetic ions, which are commonly seen in, e.g., triangle, kagome, or pyrochlore lattices. Two-dimensional magnetic kagome lattices have attracted a lot of interest in the search for QSLs [4]. Theoretically, it has been shown that the kagome system may exhibit various ordered states and different QSL ground states [5–18], such as chiral and $\mathbb{Z}_2$ QSL. These kagome QSLs are usually very close in energy [19] and depend sensitively on the particular form of the superexchange couplings, which render them difficult to be tested experimentally.

Experimental progress in finding kagome QSLs has been substantial. Among many kagome magnets, herbertsmithite ZnCu$_3$(OH)$_6$Cl$_2$ shows several promising properties of a QSL [2]. First of all, it consists of perfect kagome Cu^{2+}

($s = 1/2$) planes that show no magnetic order down to at least 20 mK [20–23]. Inelastic neutron-scattering (INS) experiments display broad dispersionless magnetic excitations that are consistent with spinon continuum expected in QSLs [24]. Later nuclear magnetic resonance (NMR) and INS experiments suggest that the system may be gapped [25,26]. Interestingly, previous studies have suggested that herbertsmithite may be close to a quantum critical point (QCP) [27]. However, it is later found that the low-energy spin excitations (< 1 meV) are dominated by the so-called "impurities" of residual interlayer Cu^{2+} ions due to imperfect substitution of interkagome Cu by Zn [26,28]. Moreover, Cu$_4$(OH)$_6$Cl$_2$, the base material that leads to herbertsmithite, has at least four polymorphs with different nuclear structures that are all different from herbertsmithite and have different magnetic orders [29–39]. INS experiments also do not support the presence of a QCP in the Zn$_x$Cu$_{4-x}$(OD)$_6$Cl$_2$ system since the antiferromagnetic (AF) order in the $x = 0$ sample becomes spin-glass-like with increasing x before the QSL is established in the $x = 1$ sample [37].

Recently, a new compound of Cu$_3$Zn(OH)$_6$FBr has been synthesized to exhibit properties that are consistent with a $\mathbb{Z}_2$ QSL [40–42]. This compound is obtained by substituting interlayer Cu^{2+} in barlowite Cu$_4$(OH)$_6$FBr with nonmagnetic

————————
*ygshi@iphy.ac.cn
†slli@iphy.ac.cn

Zn^{2+}. The barlowite has perfect Cu^{2+} kagome planes with an AF transition at about 15 K [43–46]. Since the barlowite and $Cu_3Zn(OH)_6FBr$ have the same space group for the crystal structures at room temperature, it may provide us a rare opportunity to study the quantum phase transition from an AF ordered state to a QSL ground state by continuously tuning Zn substitution level. Therefore, the $Cu_{4-x}Zn_x(OH)_6FBr$ system may provide the long-thought experimental playground for the investigation of novel quantum phase transitions between symmetry-breaking phases and symmetric topologically ordered phases.

In this paper, we systematically investigate the magnetic properties of the $Cu_{4-x}Zn_x(OH)_6FBr$ system. Our results suggest that Zn can indeed be continuously doped into barlowite and suppress the long-range AF order. However, a hexagonal-to-orthorhombic structural change is observed in barlowite, and the low-temperature magnetic structure is directly associated with the orthorhombic structure. According to x-ray diffraction and magnetic susceptibility measurements, we find that the interlayer Cu^{2+} ions result in lattice distortion and a local magnetic order up to $x = 0.82$. However, the specific-heat measurements suggest that the bulk three-dimensional (3D) antiferromagnetic order should disappear around $x = 0.4$ and the spin dynamics of QSL kagome planes may already start to evolve from $x > 0.3$, which finally leads to a gapped QSL in $Cu_3Zn(OH)_6FBr$. Putting these information together, our comprehensive results suggest that the $Cu_{4-x}Zn_x(OH)_6FBr$ system host rich physics of the interplay of frustration, antiferromagnetic order, as well as the topologically ordered QSL.

II. EXPERIMENTS

Powders of $Cu_{4-x}Zn_x(OH)_6FBr$ were synthesized by a hydrothermal method. We sealed powders of $Cu_2(OH)_2CO_3$ (1.5 mmol), NH_4F (1 mmol), $ZnBr_2$ (x_{nom} mmol), and $CuBr_2$ ($1 - x_{nom}$ mmol) in a 50-ml reaction vessel with half water, where x_{nom} is the nominal Zn content. The vessel was slowly heated to $200\,°C$ and kept for 12 h before cooling down to room temperature. Powders of $Cu_{4-x}Zn_x(OH)_6FBr$ were obtained by drying the products. The Zn content is determined by the inductively coupled plasma mass spectrometer (Thermo IRIS Intrepid II) with an uncertainty of about 4%. It should be noted for the $Cu_3Zn(OH)_6FBr$, the Zn content is determined to be about 0.92. We will use the actual Zn substitution level throughout the paper, so the $x = 0.92$ sample is equivalent to $Cu_3Zn(OH)_6FBr$ reported previously [40,42]. To obtain the deuterated samples, the mixture is changed to CuO, ZnF_2, $ZnBr_2$, and $CuBr_2$, and we use heavy water instead. The content of H is less than 2% according to the NMR measurement. The magnetic susceptibility and heat capacity were measured by the magnetic property measurement system (MPMS) and physical property measurement system (PPMS) (Quantum Design), respectively. The structures of the $Cu_{4-x}Zn_x(OH)_6FBr$ system were measured at room temperature by the x-ray diffractometer (Rigaku Ultima IV) with Cu $K\alpha$ radiation and a scintillation counterdetector. The magnetic and nuclear structures of $Cu_4(OD)_6FBr$ are determined by neutron-diffraction experiments performed
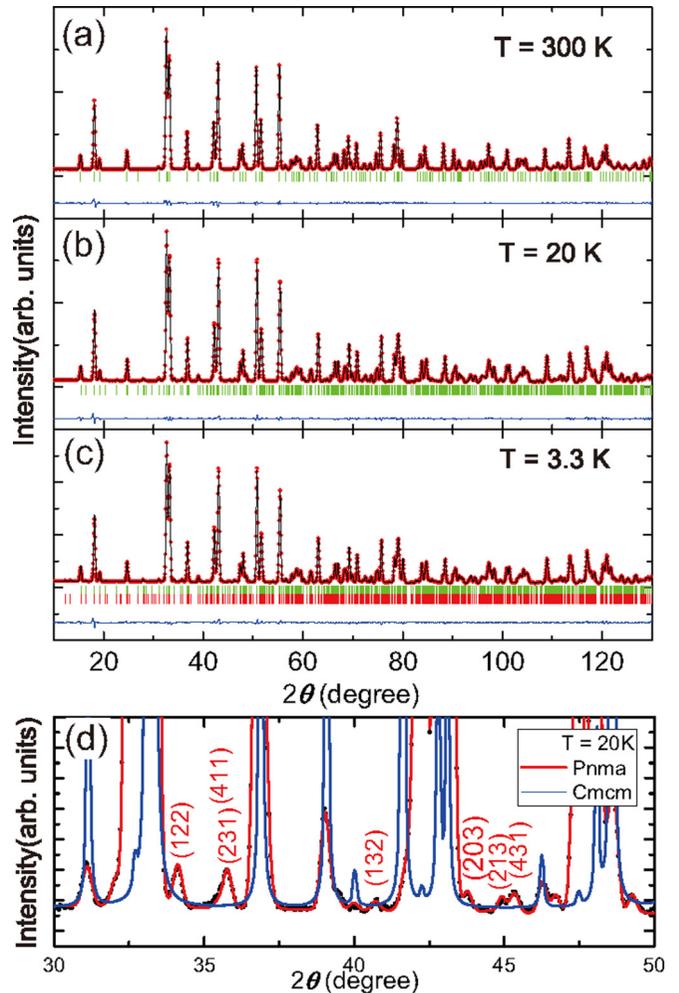

FIG. 1. Neutron powder-diffraction intensities of $Cu_4(OD)_6FBr$ (red dots) at (a) 300 K, (b) 20 K, and (c) 3.3 K measured at SPODI. The calculated intensities are shown by the black lines. Short vertical green and red lines represent nuclear and magnetic Bragg peak positions, respectively. The blue line shows the difference between measured and calculated intensities. The weighted profile R factor (R_{wp}) is 4.22%, 4.46%, and 4.29% for (a)–(c), respectively. (d) Comparison between the calculated intensities of the *Pnma* (red) and *Cmcm* (blue) structures at 20 K. The labeled peaks are in the *Pnma* notation and cannot be indexed in the *Cmcm* structure.

on the SPODI diffractometer at FRM-II, Germany and the HB-2A diffractometer at HFIR, USA with wavelengths of 1.5483 and 2.4103 Å, respectively.

III. RESULTS

A. Nuclear and magnetic structures in $Cu_4(OH)_6FBr$

Figures 1(a)–1(c) show the neutron-diffraction and refinement results of $Cu_4(OD)_6FBr$ at 300, 20, and 3.3 K, respectively [47]. The refinement of the data at 3.3 K has to include an AF structure as discussed later in this section. The nuclear structure at room temperature is found to be the same as reported previously [43] with parameters shown in Table I(a). At 20 K that is above T_N, the data can be well described by an orthorhombic structure (*Pnma*) as shown in

TABLE I. Nuclear structure parameters of $Cu_4(OD)_6FBr$. (a) At 300 K, $P6_3/mmc$ (No. 194): $a = b = 6.6760(7)$ Å, $c = 9.2952(13)$ Å, $\alpha = \beta = 90°$, $\gamma = 120°$. Rp:4.25%, Rwp:4.22%, and χ^2:7.55. (b) At 20 K, $Pnma$ (No. 62): $a = 11.5129(42)$ Å, $b = 9.2703(33)$ Å, $c = 6.6801(28)$ Å, $\alpha = \beta = \gamma = 90°$. Rp:3.50%, Rwp:4.40%, and χ^2:13.2.

(a)	Site	x	y	z	B (Å^2)
Cu1	6g	0.00000	0.50000	0.00000	1.403(2)
Cu2	6h	0.36993	0.63007(15)	0.750000	1.089(5)
Br	2c	0.66667	0.33333	0.75000	1.788(3)
F	2b	0.00000	0.00000	0.750000	2.137(4)
O	12k	0.20234	0.79766(7)	0.90850(7)	1.271(2)
D	12k	0.13470	0.87530(7)	0.86677(8)	2.235(2)
(b)	Site	x	y	z	B (Å^2)
Cu1	4a	0.00000	0.00000	0.00000	0.784(3)
Cu2	8d	0.25076(21)	0.51109(17)	0.24514(28)	0.728(2)
Cu3	4c	0.18593(19)	0.25000	0.05766(24)	0.856(3)
Br	4c	0.33072(23)	0.25000	0.50431(50)	0.766(2)
F	4c	0.49753(30)	0.25000	0.00719(70)	1.107(3)
O1	8d	0.29741(16)	0.09595(23)	0.00115(45)	0.753(3)
O2	8d	0.10228(24)	0.09210(28)	0.19800(41)	0.772(4)
O3	8d	0.40066(25)	0.58780(26)	0.30230(39)	0.755(4)
D1	8d	0.37649(15)	0.13516(26)	0.00117(39)	1.543(4)
D2	8d	0.06257(25)	0.13019(26)	0.31516(39)	1.619(5)

Table I(b). Recently, an orthorhombic structure with a different space-group $Cmcm$ was reported for the single-crystal barlowite [48]. Figure 1(d) shows the comparison between the calculated results of these two structures at 20 K on our sample where the $Cmcm$ structure cannot describe the data. This discrepancy may come from the different methods used to prepare the samples.

Figure 2(a) depicts the change in the structure in the view vertical to the kagome planes. According to the low-temperature structure, we label the three Cu^{2+} ions in the kagome planes as Cu(1), Cu(2a), and Cu(2b) and the inter-layer Cu^{2+} as Cu(3). At room temperature, the first three of them are symmetrically equivalent [Cu1 in Table I(a)] and forms an equilateral triangle. Cu(3) [Cu2 in Table I(a)] has three equivalent positions with an average position at the center of the triangle as shown in the left panel of Fig. 2(a). The distribution of Cu(3) ions is presumably random. At 20 K, this random distribution is replaced by a regular pattern as shown in the right panel of Fig. 2(a), and the position of Cu3 is shown in Table I(b). In the meantime, the positions of Cu(2a) and Cu(2b), which are symmetrically equivalent in the $Pnma$ space group and labeled as Cu2 in Table I(b), are distorted both within and out of the kagome plane whereas that of Cu(1) [Cu1 in Table I(b)] remains unchanged. The new unit cell is defined according to the positions of the Cu(1) ions.

Figure 2(b) gives the values of the Cu-O bond lengths and Cu-O-Cu angles at 20 K. According to the Goodenough-Kanamori rule [49,50], the nearest-neighbor superexchange changes from positive (antiferromagnetic) to negative (ferromagnetic) when the Cu-O-Cu angle goes through about 95° [51]. The Cu-O-Cu angles among Cu(3) and three other Cu^{2+} have values both larger and smaller than 95° at 20 K, suggesting very complicated superexchange couplings.
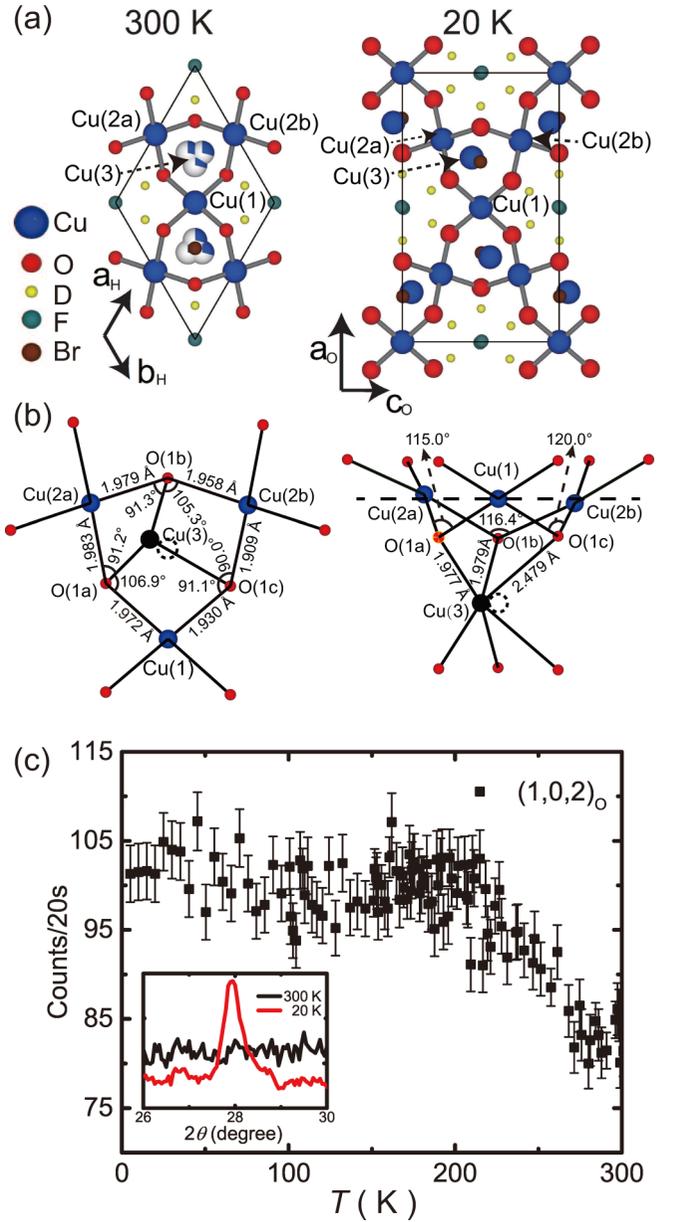

FIG. 2. (a) Nuclear structure at 300 K (left) and 20 K (right) in the view vertical to the kagome plane. The diamond and rectangle are in-plane unit cells at 300 and 20 K, respectively. The subscripts of "H" and "O" denote hexagonal and orthorhombic structures, respectively. We note that the c axis at room temperature becomes the b axis at low temperatures. The three overlapping Cu atoms at 300 K represent three positions that the atom may actually occupy. (b) Detailed low-temperature structure showing only Cu (large blue circles) and O (small red circles) in the view parallel to orthorhombic b (left) and orthorhombic a (right), respectively. (c) Temperature dependence of the $(1, 0, 2)_O$ structural peak measured at HB-2A. The inset shows 2θ scans around this peak at 20 and 300 K.

Figure 2(c) shows the temperature dependence of the intensity of the $(1, 0, 2)_O$ structural peak in the orthorhombic notation, which appears below about 270 K and slowly increases with decreasing temperature until about 200 K. As discussed above, the structural change from high temperature to low temperature is associated with a particular

position chosen by Cu(3) from three equivalent positions. Since the position of Cu(3) is randomly distributed among these three positions at room temperature, it follows that Cu(3) ions should be able to resonate among them above 200 K so that their positions are not random anymore at low temperatures. At the current stage due to the lack of thermodynamic evidence, we are unable to distinguish whether this structural change is a phase transition or rather a crossover resulting from increasing distortion with decreasing temperature.

Assuming that there is no structural transition below 20 K, we find that the neutron-diffraction data at 3.3 K [Fig. 1(c)] can be refined by introducing an AF order. To be more explicit, the inset of Fig. 3(a) shows the neutron-diffraction data and the refinement results for the first three peaks with lowest angles. Two new peaks at $(110)_O$ and $(001)_O$ emerge at 3.3 K, which are forbidden in the orthorhombic structure. The intensity of the $(101)_O$ peak significantly increases from 20 to 3.3 K. The main panel of Fig. 3(a) gives the subtracted results, which clearly shows three magnetic peaks. Figure 3(b) shows the temperature dependence of the intensities of two peaks, which is consistent with the AF transition at about 15 K reported previously [43], suggesting their magnetic origin. Within the statistics, only one magnetic transition is observed. With T_N fixed at 15.5 K determined by the specific-heat measurements as shown in the next section, the intensities of the $(0, 0, 1)_M$ and $(1, 0, 1)_M$ AF Bragg peaks can be fitted by $A(1 - T/T_N)^{2\beta}$ with β as 0.39 ± 0.06 and 0.33 ± 0.04, respectively. These values of the order parameter critical exponent are consistent with the 3D Heisenberg universality [52].

With only three magnetic peaks observed, the magnetic structure cannot be unambiguously determined. Symmetry analysis of the different possible magnetic structures is performed using SARAh [53], which gives the propagation vector $k_{19} = (0, 0, 0)$. Representational analysis on three copper sites gives eight irreducible representations (IRs) where IR-7 gives the best refinement results. After trying different models, the following conclusions can be made. First, the magnetic unit cell is the same as the orthorhombic one. Therefore, the positions of the $(110)_O$, $(001)_O$, and $(101)_O$ peaks in the inset of Fig. 3(a) are the same as those of the $(110)_M$, $(001)_M$, and $(101)_M$ peaks in the main panel of Fig. 3(a). Second, the moments on all copper sites are always confined within kagome planes, i.e., the ac plane in the orthorhombic structure. Third, the magnetic structures of the Cu(3) ions are always the same with the moment direction along the orthorhombic *a* axis as shown in Fig. 3(c). The magnetic moment is about $0.66(7)\mu_B$, which changes little in different models.

Although the magnetic configurations of the Cu(1) and Cu(2) ions cannot be determined, we give three examples of refinement results as shown by the solid lines in Fig. 3(a). In the $M1$ configuration, the moments at the Cu(1) and Cu(2) ions are fixed to be the same, and their directions rotate simultaneously. The value of the moments is found to be about $0.182(35)\mu_B$ with the moment direction about $30°$ away from the orthorhombic *a* axis. In the $M2$ configuration, the moment at the Cu(2) ions is set to zero. The value of the moment is about $0.39(22)\mu_B$, but the direction is rather arbitrary. In the $M3$ configuration, the directions of the Cu(1) and Cu(2)
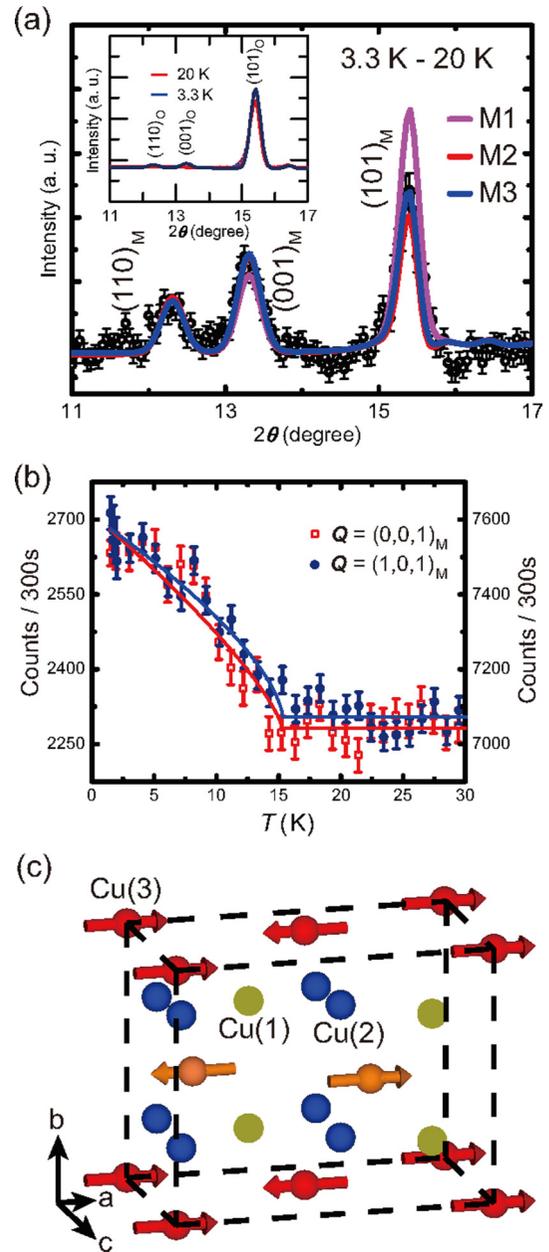

FIG. 3. (a) The three magnetic peaks obtained by subtracting the 20-K data from the 3.3-K data. The solid lines are calculated results of three types of magnetic structures. The inset shows the refined results of raw data measured at SPODI. The subscript "*M*" denotes the magnetic structure. (b) Temperature dependence of $(0, 0, 1)_M$ (open red squares) and $(1, 0, 1)_M$ (filled blue circles) magnetic peaks measured at HB-2A. The solid lines are the fitted results as described in the main text. (c) Magnetic structure with the arrows indicating the sizes and directions of magnetic moments on Cu(3) ions (red and orange balls). The yellow and blue balls are Cu(1) and Cu(2) ions, respectively, whose structures cannot be determined as discussed in the main text. The dashed black lines indicate the magnetic unit cell with the origin shifted.

moments are set to be along the orthorhombic *a* axis, the same as that of the Cu(3) moment. The values of the moments on the Cu(1) and Cu(2) ions are $0.329(76)\mu_B$ and $0.079(37)\mu_B$, respectively. Although $M3$ gives the best fit, it cannot be really
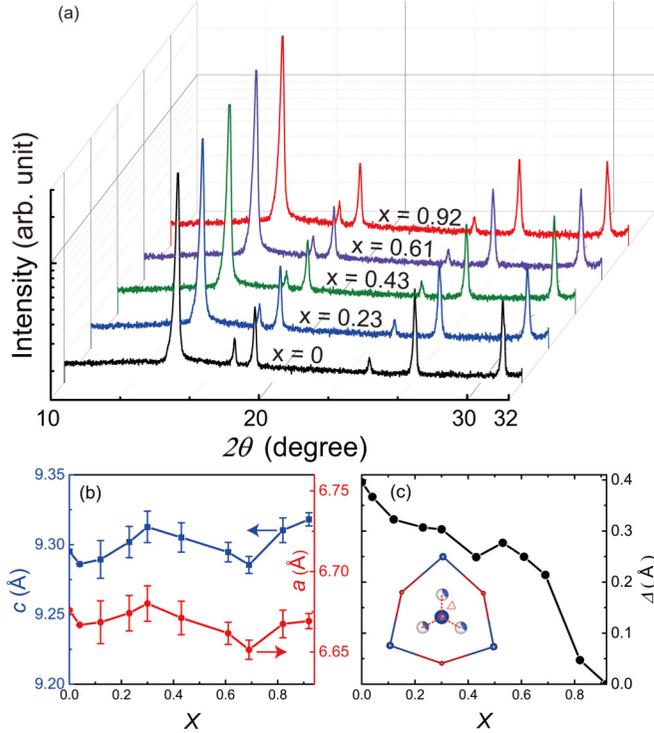
FIG. 4. (a) Room-temperature x-ray-diffraction data of selected $Cu_{4-x}Zn_x(OH)_6FBr$ samples. The intensity axis is plotted with the logarithmic scale. (b) Evolution of lattice constants with Zn substitution level x at room temperature. We note that $a_H = b_H$. (c) Doping dependence of the splitting Δ of Cu(3) from its average position as defined in the inset.

distinguished from the other configurations within the error bars. In any case, the average moment of Cu(1) and Cu(2) is much smaller than that of Cu(3), which is why the magnetic structure of Cu(3) ions can be settled.

B. Evolution of the antiferromagnetic order with Zn substitution

Figure 4(a) shows the room-temperature x-ray-diffraction data of $Cu_{4-x}Zn_x(OH)_6FBr$ with different x's from 0 to 0.92. Since there is no new peak appearing with Zn substitution, one can conclude that Zn can be continuously doped into barlowite. All the data can be refined by the hexagonal structure with the space group of $P6_3/mmc$. Figure 4(b) gives the substitution evolution of lattice constants a_H and c_H, both of which change little with substitution. As discussed in the previous subsection, the interlayer Cu^{2+} in barlowite has to be refined with three equivalent positions. One can define Δ as the distance from one of these positions to the center of their average positions as shown by the inset of Fig. 4(c). The value of Δ continuously decreases with increasing x and becomes zero in the $x = 0.92$ sample as shown in Fig. 4(c). The Cu-O-Cu angle between Zn^{2+} or Cu^{2+} at the Cu(3) position and other Cu^{2+} ions on kagome planes of the $x = 0.92$ sample is $94.73°$ at 4 K according to our previous measurements [42], which will give nearly zero superexchange couplings [51]. Therefore, it seems that the presence of residual interlayer Cu^{2+} will not affect the spin dynamics of the kagome layers as far as its content is less than 10%. It should be noted the above
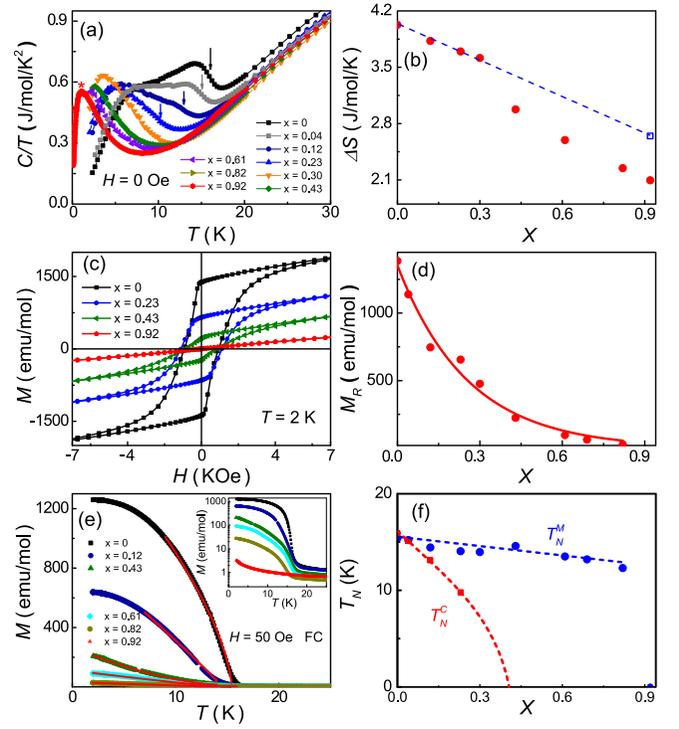

FIG. 5. (a) Low-temperature specific heat of $Cu_{4-x}Zn(OH)_6FBr$. All the samples were measured down to 2 K except for the $x = 0.92$ sample, which has already been reported previously [40]. The arrows indicate the bulk magnetic transition temperature T_N, which is determined as the middle point of the drop of $\Delta C/T$ during the transition. The low-temperature hump for the $x = 0.92$ sample is indicated by the star symbol. (b) Temperature dependence of ΔS as defined in the main text. The red filled circles are results considering C/T above 2 K, whereas the temperature in obtaining the data of the blue open squares is down to 50 mK. (c) Field dependence of magnetization at 2 K for selected samples at low fields. (d) Doping dependence of retentivity M_R. The solid line is an exponential fit. (e) Low-temperature magnetic moments of $Cu_{4-x}Zn_x(OH)_6FBr$ measured by the field-cooling process at 50 Oe. The solid lines are fitted results as described in the main text. The inset shows the data with the logarithmic scale. (f) Doping dependence of T_N from specific heat (T_N^C, red squares) and susceptibility (T_N^M, blue circles) measurements. The T_N of the $x = 0.92$ sample is manually set as zero. The dashed lines are guides to the eye.

analysis has assumed that Zn^{2+} only substitutes interlayer Cu^{2+}. At the current stage, we cannot exclude the possibility that a few amounts of Zn^{2+} may substitute Cu^{2+} within the kagome planes [25,54,55].

The AF transition in barlowite can be observed in specific heat as shown in Fig. 5(a), which is similar as reported previously [43]. In the $x = 0.92$ sample, a hump is found in C/T as indicated by the star symbol in Fig. 5(a) [40], which has been attributed to the contribution from residual interlayer Cu^{2+} in herbertsmithite [26]. With decreasing Zn content, the hump does not disappear but moves to higher temperatures with little change in the integrated area. On the other hand, starting from the $x = 0$ sample, the AF transition can be still seen in the $x = 0.12$ sample, but becomes indistinguishable due to the presence of the hump at low temperatures for $x \geqslant 0.3$.

We may estimate the contribution of this hump by analyzing the entropy change ΔS during the magnetic transition. The high-temperature data are fitted by a simple polynomial function $C_{bg} = \alpha T^2 + \beta T^3$ as performed previously [43]. The range has been chosen from 30 K to about 5 K higher than T_N for $x < 0.23$ or 15 K for others. The magnetic part of the specific-heat C_M associated with the AF order can be obtained by subtracting C_{bg} from the raw data. It should be noted that a substantial magnetic contribution to the specific heat should be present above T_N as discussed in the following subsection and the quadratic term in C_{bg} may be associated with it [43]. ΔS can thus be obtained by integrating C_M/T from 0 to 20 K, assuming that C_M/T $(T = 0\,\mathrm{K}) = 0$, which is the case for either the AF ordered or the QSL ground state. However, we have only measured the specific heat down to 2 K for most of the samples as shown in Fig. 5(a). The above process will significantly underestimate the contribution below 2 K for samples with large x since the hump temperature becomes lower than 2 K. We have measured the specific heat of the $x = 0.92$ sample down to 50 mK, which can give us a more precise value of ΔS as shown by the blue square in Fig. 5(b). A rough linear substitution dependence of ΔS from the low-substitution samples to the $x = 0.92$ sample is found, suggesting that the hump contribution of ΔS is rather independent of substitution. Therefore, the actual entropy change in ΔS in barlowite is just about 1.38 J/mol/K, which corresponds to $0.06 k_B \ln 2$ per Cu^{2+}, only about one-third of that reported previously [43].

Figure 5(c) shows low-temperature field dependence of magnetization M at 2 K. Ferromagneticlike hysteresis can be found in all the samples except for the $x = 0.92$ one. The substitution dependence of retentivity M_R, i.e., the magnetization at the zero field after the magnetic field is removed, is shown in Fig. 5(d), where M_R decreases exponentially with increasing x. The presence of ferromagneticlike hysteresis most likely comes from the domains formed at low temperatures due to the orthorhombic structure. The exponential decrease in M_R suggests that either the energy required for overcoming the domain walls or the ferromagneticlike component of the bulk order decreases quickly with increasing x.

Figure 5(e) shows the temperature dependence of magnetic moment M below 25 K at 50 Oe. The low-temperature signal decreases dramatically with increasing x, but we can still observe magnetic transitions for samples with x up to 0.82. Fitting the data with $A(1 - T/T_N)^\beta$ gives the substitution dependence of T_N^M in Fig. 5(f), where T_N^M only decreases slightly with increasing x and suddenly becomes zero at $x = 0.92$. On the other hand, T_N^C, obtained from the specific-heat measurements in Fig. 5(a), decreases quickly with Zn substitution and may become zero around $x = 0.4$. It should be pointed out that this value is very subjective and since we cannot observe T_N^C around $x = 0.4$. Further studies are thus needed to identify the exact substitution level.

C. High-temperature and high-field properties of $Cu_{4-x}Zn_x(OH)_6FBr$

Figure 6(a) shows the temperature dependence of $1/\chi$ of $Cu_{4-x}Zn_x(OH)_6FBr$. The high-temperature data from 150 and 300 K can be fitted by the Curie-Weiss function as
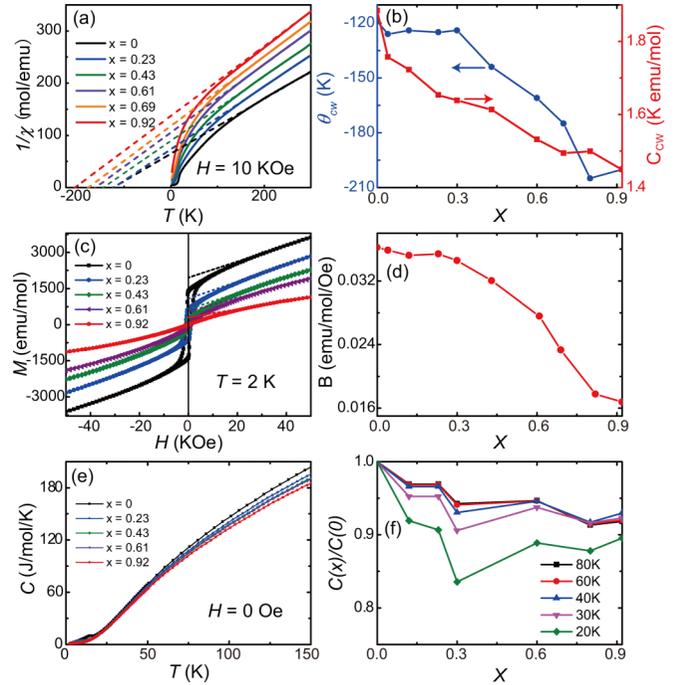

FIG. 6. (a) Temperature dependence of $1/\chi$ for $Cu_{4-x}Zn_x(OH)_6FBr$. The dashed lines are fitted results between 150 and 300 K. (b) Doping dependence of the Curie temperature θ_{CW} (left axis) and Curie constant C_{CW} (right axis). (c) Field dependence of magnetization at 2 K for selected samples at high fields. The dashed lines are linear fitted results from 30 to 50 KOe. (d) Doping dependence of slope B fitted from (c). (e) High-temperature specific heat of selected samples. (f) Doping dependence of $C(x)/C(0)$ at several temperatures.

$\chi = C_{CW}/(T - \theta_{CW})$. The Curie constant C_{CW} decreases linearly with increasing substitution as shown in Fig. 6(b), which suggests that it is associated with the content of Cu^{2+} ions. However, the absolute value of the Curie temperature θ_{CW} only starts increasing above $x = 0.3$, which is consistent with what is observed in the $Zn_xCu_{4-x}(OH)_6Cl_2$ system [20]. This implies there is a substantial change in the nature of magnetic interactions in the system from $x > 0.3$, and we believe it is related to the appearance of the magnetic properties of the kagome QSL plane from $x > 0.3$ onwards.

Figure 6(c) shows high-field magnetization at 2 K. The data between 30 and 50 KOe can be fitted by a linear function as $A + BH$. We note that similar measurements on the single crystal of barlowite have shown large anisotropy for fields parallel and perpendicular to the c axis, but the slopes in the above field range are rather the same [45]. Figure 6(d) shows substitution dependence of B, which starts decreasing above $x = 0.3$. This is consistent with the strengthening of antiferromagnetic correlations as indicated by the increasing in $|\theta_{CW}|$.

Figure 6(e) shows the specific-heat data up to 150 K. Apart from low-temperature differences due to the presence of the AF order and hump [Fig. 5(e)], the high-temperature data also show different behaviors. Figure 6(f) plots the substitution dependence of $C(x)/C(0)$, where $C(x)$ and $C(0)$ are the specific heat with Zn substitution levels of x and zero, respec-

tively. This value roughly decreases monotonically with x for $T > 60$ K, which suggests that the high-temperature specific heat may be dominated by phonons. However, a dip is found at $x = 0.3$ for those at lower temperatures, indicating that there is a contribution from the kagome QSL plane as suggested in Figs. 6(b) and 6(d). This is also consistent with the observation that only a small amount of entropy is involved during the magnetic transition in barlowite as shown in the previous section.

IV. DISCUSSIONS

Our results provide a comprehensive picture of the magnetic order in barlowite. The establishment of the AF order is directly associated with the structural distortion at high temperatures, which is the reason that the magnetic structure in Fig. 3 is different from the incorrectly proposed canted antiferromagnetic order based on the first-principles calculation [44] without knowing the structural change. It is interesting to note that three different magnetic structures have been proposed for the clinoatacamite $Cu_2(OH)_3Cl$ [37–39]. In our case, the magnetic structure in the $M3$ configuration is consistent with results from bulk measurements on barlowite. For example, the magnetic entropy associated with the magnetic transition is just about $0.06 k_B \ln 2$ per Cu^{2+}. In the simplest model, the magnetic entropy is proportional to M^2, i.e., the square of the magnetic moment [56]. Here the sum of the ordered moments of four Cu^{2+} ions in the $M3$ configuration is $1.076 \mu_B$, whereas the total moment is $4gS = 4.54 \mu_B$, taking $g = 2.27$ and $S = 1/2$ [43]. Therefore, the entropy release above the transition is about 5.6% of $k_B \ln 2$, which is very close to the experimental value. Moreover, in the magnetic measurements on the single crystal, the hysteresis loop is only observed when $H \perp c$ but not for $H \parallel c$ as shown in the single-crystal measurements [45], which is consistent with our results that the magnetic/structural domains only present within the kagome planes. When the magnetic field is parallel to the c axis, a saturation moment of $0.29 \mu_B$ per Cu is found [45]. Although it is attributed to full polarization of the interlayer Cu^{2+}, we find that this value is close to the sum of the ordered moments in the $M3$ configuration.

The presence of interlayer Cu^{2+} ions may result in local lattice distortions that will give rise to the magnetic order even when the Zn substitution level is high. This picture is consistent with our observation of magnetic order in magnetic measurements and low-field hysteresis up to $x = 0.82$, almost substitution independent of T_N^M. It also coincides with the nuclear structure refinement results at room temperature, which suggests that the splitting of Δ of Cu(3) is not zero even when x is as large as 0.82. However, the results from the specific-heat measurements provide another picture where the magnetic order may have already become zero for x larger than 0.4. Since specific heat is a bulk property, it suggests that the magnetic order established at T_N^M is just a secondary phase. This is consistent with our high-temperature and high-magnetic-field results, which suggest that the spin dynamics of the QSL kagome plane may indeed start to appear with Zn substitution level $x > 0.3$.

Based on these analyses, the magnetic properties in the $Cu_{4−x}Zn_x(OH)_6FBr$ system can be divided into two parts,

the one associated with the kagome planes (and thus bulk) and the one associated with interlayer Cu^{2+} moments (and thus local). In very low substitution samples, the two parts are strongly coupled and cannot be separated, hence the 3D antiferromagnetic order is formed. With increasing Zn substitution, the bulk magnetic order is quickly suppressed and may disappear around $x \sim 0.4$, but the local magnetic order persists up to $x = 0.82$ without much change in its T_N. It is worth noting that in both the herbertsmithite and the $x = 0.92$ samples, the spin excitations can be indeed separated into these two independent parts [26,28,42].

The suppression of bulk magnetic order gives rise to two possible scenarios. In the first one, a magnetic QCP is present around $x \sim 0.4$, which suggests that $Cu_{4−x}Zn_x(OH)_6FBr$ may provide us the long-thought opportunity to study the quantum phase transition from a magnetic ordered state to a QSL state. On the other hand, it is also possible that the $\mathbb{Z}_2$ QSL state in $Cu_3Zn_x(OH)_6FBr$ is very robust against interlayer magnetic impurities so that it may persists up to very low Zn substitution. In this case, the disappearance of the bulk magnetic order may be associated with a first-order quantum phase transition or even phase separation between the 3D AF order and the QSL. It should be noted that since the $x = 0$ and $x = 0.92$ samples have different structures at low temperatures, it will be interesting to see whether the quantum transition around $x = 0.4$ happens within the same nuclear structure or not. In any case, the physics is rich and interesting, and further studies are needed to clarify the situation.

V. CONCLUSIONS

Our systematical investigation on the Zn substitution effect on the antiferromagnetism in the kagome $Cu_{4−x}Zn_x(OH)_6FBr$ system have revealed three major conclusions. First, the magnetic order in barlowite is associated with a hexagonal-to-orthorhombic structural change. Second, Zn substitution leads to local lattice distortion and may give rise to phase separation and result in a bulk magnetic order and a local magnetic order. Third, an evolution of spin dynamics on the kagome QSL planes may result in a quantum phase transition around $x = 0.4$ between 3D AF order and QSL. Our results suggest that $Cu_{4−x}Zn_x(OH)_6FBr$ is an interesting system and an experimental playground to investigate the intriguing physics of kagome antiferromagnets and possibly realize the long-thought situation where quantum phase transitions between symmetry-breaking and topologically ordered phases. Further works are definitely needed to explore the rich physics in these kagome compounds.

[1] L. Balents, Nature (London) **464**, 199 (2010).

[2] M. R. Norman, Rev. Mod. Phys. **88**, 041002 (2016).

[3] L. Savary and L. Balents, Rep. Prog. Phys. **80**, 016502 (2017).

[4] P. A. Lee, Science **321**, 1306 (2008).

[5] S. Sachdev, Phys. Rev. B **45**, 12377 (1992).

[6] H. C. Jiang, Z. Y. Weng, and D. N. Sheng, Phys. Rev. Lett. **101**, 117203 (2008).

[7] S. Yan, D. A. Huse, and S. R. White, Science **332**, 1173 (2011).

[8] H.-C. Jiang, Z. Wang, and L. Balents, Nat. Phys. **8**, 902 (2012).

[9] L. Messio, B. Bernu, and C. Lhuillier, Phys. Rev. Lett. **108**, 207204 (2012).

[10] M. Punk, D. Chowdhury, and S. Sachdev, Nat. Phys. **10**, 289 (2014).

[11] S. Bieri, L. Messio, B. Bernu, and C. Lhuillier, Phys. Rev. B **92**, 060407 (2015).

[12] Y. Iqbal, H. O. Jeschke, J. Reuther, R. Valentí, I. I. Mazin, M. Greiter, and R. Thomale, Phys. Rev. B **92**, 220404 (2015).

[13] K. Kumar, K. Sun, and E. Fradkin, Phys. Rev. B **92**, 094433 (2015).

[14] S.-S. Gong, W. Zhu, K. Yang, O. A. Starykh, D. N. Sheng, and L. Balents, Phys. Rev. B **94**, 035154 (2016).

[15] H. J. Liao, Z. Y. Xie, J. Chen, Z. Y. Liu, H. D. Xie, R. Z. Huang, B. Normand, and T. Xiang, Phys. Rev. Lett. **118**, 137202 (2017).

[16] J.-W. Mei, J.-Y. Chen, H. He, and X.-G. Wen, Phys. Rev. B **95**, 235107 (2017).

[17] Y.-C. Wang, C. Fang, M. Cheng, Y. Qi, and Z. Y. Meng, arXiv:1701.01552 .

[18] Y.-C. Wang, X.-F. Zhang, F. Pollmann, M. Cheng, and Z. Y. Meng, Phys. Rev. Lett. **121**, 057202 (2018).

[19] P. Mendels and F. Bert, C. R. Phys. **17**, 455 (2016).

[20] M. P. Shores, E. A. Nytko, B. M. Bartlett, and D. G. Nocera, J. Am. Chem. Soc. **127**, 13462 (2005).

[21] F. Bert, S. Nakamae, F. Ladieu, D. L'Hôte, P. Bonville, F. Duc, J.-C. Trombe, and P. Mendels, Phys. Rev. B **76**, 132411 (2007).

[22] P. Mendels, F. Bert, M. A. de Vries, A. Olariu, A. Harrison, F. Duc, J. C. Trombe, J. S. Lord, A. Amato, and C. Baines, Phys. Rev. Lett. **98**, 077204 (2007).

[23] J. S. Helton, K. Matan, M. P. Shores, E. A. Nytko, B. M. Bartlett, Y. Yoshida, Y. Takano, A. Suslov, Y. Qiu, J.-H. Chung, D. G. Nocera, and Y. S. Lee, Phys. Rev. Lett. **98**, 107204 (2007).

[24] T.-H. Han, J. S. Helton, S. Chu, D. G. Nocera, J. A. Rodriguez-Rivera, C. Broholm, and Y. S. Lee, Nature (London) **492**, 406 (2012).

[25] M. Fu, T. Imai, T.-H. Han, and Y. S. Lee, Science **350**, 655 (2015).

[26] T.-H. Han, M. R. Norman, J.-J. Wen, J. A. Rodriguez-Rivera, J. S. Helton, C. Broholm, and Y. S. Lee, Phys. Rev. B **94**, 060409(R) (2016).

[27] J. S. Helton, K. Matan, M. P. Shores, E. A. Nytko, B. M. Bartlett, Y. Qiu, D. G. Nocera, and Y. S. Lee, Phys. Rev. Lett. **104**, 147201 (2010).

[28] G. J. Nilsen, M. A. de Vries, J. R. Stewart, A. Harrison, and H. M. Rønnow, J. Phys. Condens. Matter **25**, 106001 (2013).

[29] F. C. Hawthorne, Mineral Mag. **49**, 87 (1985).

[30] J. B. Parise and B. G. Hyde, Acta Crystallogr., Sect. C: Cryst. Struct. Commun. **42**, 1277 (1986).

[31] J. D. Grice, J. T. Szymanski, and J. L. Jambor, Can. Mineral. **34**, 73 (1996).

[32] J. L. Jambor, J. E. Dutrizac, A. C. Roberts, J. D. Grice, and J. T. Szymanski, Can. Mineral. **34**, 61 (1996).

[33] T. Malcherek and J. Schluter, Acta Crystallogr., Sect. B: Struct. Sci. **65**, 334 (2009).

[34] X. G. Zheng, T. Mori, K. Nishiyama, W. Higemoto, H. Yamada, K. Nishikubo, and C. N. Xu, Phys. Rev. B **71**, 174404 (2005).

[35] X. G. Zheng, T. Kawae, Y. Kashitani, C. S. Li, N. Tateiwa, K. Takeda, H. Yamada, C. N. Xu, and Y. Ren, Phys. Rev. B **71**, 052409 (2005).

[36] X. G. Zheng, H. Kubozono, K. Nishiyama, W. Higemoto, T. Kawae, A. Koda, and C. N. Xu, Phys. Rev. Lett. **95**, 057201 (2005).

[37] S.-H. Lee, H. Kikuchi, Y. Qiu, B. Lake, Q. Huang, K. Habicht, and K. Kiefer, Nature Mater. **6**, 853 (2007).

[38] J.-H. Kim, S. Ji, S.-H. Lee, B. Lake, T. Yildirim, H. Nojiri, H. Kikuchi, K. Habicht, Y. Qiu, and K. Kiefer, Phys. Rev. Lett. **101**, 107201 (2008).

[39] A. S. Wills and J.-Y. Henry, J. Phys.: Condens. Matter **20**, 472206 (2008).

[40] Z. Feng, Z. Li, X. Meng, W. Yi, Y. Wei, J. Zhang, Y.-C. Wang, W. Jiang, Z. Liu, S. Li, F. Liu, J. Luo, S. Li, G.-q. Zheng, Z. Y. Meng, J.-W. Mei, and Y. Shi, Chin. Phys. Lett. **34**, 077502 (2017).

[41] X.-G. Wen, Chin. Phys. Lett. **34**, 90101 (2017).

[42] Y. Wei, Z. Feng, W. Lohstroh, C. dela Cruz, W. Yi, Z. F. Ding, J. Zhang, C. Tan, L. Shu, Y.-C. Wang, J. Luo, J.-W. Mei, Z. Y. Meng, Y. Shi, and S. Li, arXiv:1710.02991.

[43] T.-H. Han, J. Singleton, and J. A. Schlueter, Phys. Rev. Lett. **113**, 227203 (2014).

[44] H. O. Jeschke, F. Salvat-Pujol, E. Gati, N. H. Hoang, B. Wolf, M. Lang, J. A. Schlueter, and R. Valentí, Phys. Rev. B **92**, 094417 (2015).

[45] T.-H. Han, E. D. Isaacs, J. A. Schlueter, and J. Singleton, Phys. Rev. B **93**, 214416 (2016).

[46] K. M. Ranjith, C. Klein, A. A. Tsirlin, H. Rosner, C. Krellner, and M. Baenitz, Sci. Rep. **8**, 10851 (2018).

[47] See Supplemental Material at http://link.aps.org/supplemental/10.1103/PhysRevB.98.155127 for CIF files.

[48] C. M. Pasco, B. A. Trump, T. T. Tran, Z. A. Kelly, C. Hoffmann, I. Heinmaa, R. Stern, and T. M. McQueen, Phys. Rev. Mater.s **2**, 044406 (2018).

[49] J. B. Goodenough, J. Phys. Chem. Solids **6**, 287 (1958).

[50] J. Kanamori, J. Phys. Chem. Solids **10**, 87 (1959).

[51] Y. Mizuno, T. Tohyama, S. Maekawa, T. Osafune, N. Motoyama, H. Eisaki, and S. Uchida, Phys. Rev. B **57**, 5326 (1998).

[52] M. Campostrini, M. Hasenbusch, A. Pelissetto, P. Rossi, and E. Vicari, Phys. Rev. B **65**, 144520 (2002).

[53] A. S. Wills, Physica B **276**, 680 (2000).

[54] A. Olariu, P. Mendels, F. Bert, F. Duc, J. C. Trombe, M. A. de Vries, and A. Harrison, Phys. Rev. Lett. **100**, 087202 (2008).

[55] D. E. Freedman, T. H. Han, A. Prodi, P. Müller, Q.-Z. Huang, Y.-S. Chen, S. M. Webb, Y. S. Lee, T. M. McQueen, and D. G. Nocera, J. Am. Chem. Soc. **132**, 16185 (2010).

[56] F. Schwabl, *Statistical Mechanics*, 2nd ed., Advanced Texts in Physics (Springer-Verlag, Berlin/Heidelberg, 2006).